

Atomistic QM/Classical Modeling of Surface-Enhanced Infrared Absorption

Sveva Sodomaco, Piero Lafiosca, Tommaso Giovannini,* and Chiara Cappelli*

Cite This: *J. Phys. Chem. C* 2026, 130, 1919–1930

Read Online

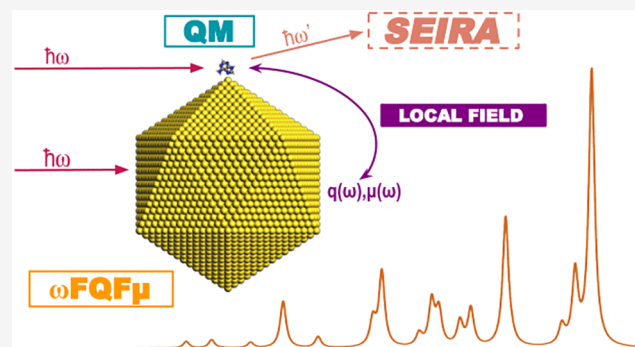
ACCESS |

Metrics & More

Article Recommendations

Supporting Information

ABSTRACT: We present a multiscale quantum mechanics/classical (QM/MM) approach for modeling surface-enhanced infrared absorption (SEIRA) spectra of molecules adsorbed on plasmonic nanostructures. The molecular subsystem is described at the density functional theory (DFT) level, while the plasmonic material is represented using fully atomistic, frequency-dependent Fluctuating Charges (ω FQ) and Fluctuating Charges and Dipoles (ω FQF μ) models. These schemes enable an accurate and computationally efficient description of the plasmonic response of both graphene-based materials and noble metal nanostructures, achieving accuracy comparable to that of *ab initio* methods. The proposed methodology is applied to the calculation of SEIRA spectra of adenine adsorbed on gold nanoparticles and graphene sheets. The quality and robustness of the approach are assessed through comparison with surface-enhanced Raman scattering (SERS) spectra and available experimental data. The results demonstrate that the proposed framework provides a reliable route to simulate vibrational responses of plasmon–molecule hybrid systems.



1. INTRODUCTION

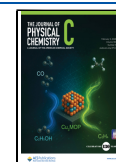
Surface-enhanced vibrational spectroscopies, including surface-enhanced infrared absorption (SEIRA) and surface-enhanced Raman scattering (SERS), selectively amplify molecular vibrational signals by exploiting the intense local and inhomogeneous electric field enhancement near plasmonic nanostructures.^{1–4} SERS is among the most widely employed techniques in (bio)sensing due to its high sensitivity to the molecular fingerprints, which can be enhanced by several orders of magnitude, even allowing single molecule detection.⁵ SEIRA has found applications in diverse fields, including the in situ monitoring of metal surface catalytic reactions,^{6,7} electrochemical studies,^{8–10} nanoparticle design,^{11,12} and analytical biosensing platforms.^{13–16}

First reported in 1980 via attenuated total reflection measurements,¹⁷ SEIRA was initially studied for molecular monolayers adsorbed on metal films (e.g., Au, Ag, Cu, Pt), and on core–shell substrates.^{18,19} These configurations typically yield enhancement factors up to 10^4 .^{17,20} More recently, the interest has shifted toward resonant SEIRA, which seeks to further improve sensitivity by engineering metal nanoantennas or graphene-based structures to make the plasmon resonance frequency (PRF) resonate with the vibrational modes of target molecules, i.e., in the infrared region.²¹ In particular, graphene and its derivatives have gained significant attention as SEIRA substrates due to their tunable plasmonic response in the mid-IR range.²² Similarly to SERS, the dominant enhancement mechanism in SEIRA arises from the local electromagnetic

(EM) mechanism at the molecule–substrate interface, which is primarily linked to the excitation of the localized surface plasmon. However, while SERS enhancement depends on the fourth power of the local field, SEIRA exhibits a quadratic dependence, yielding intrinsically lower intensities as compared to SERS.¹

Theoretical modeling plays a crucial role in predicting and interpreting SEIRA and SERS spectra.¹⁹ Full Quantum Mechanical (QM) approaches have typically considered molecules adsorbed on small noble metal clusters.^{14,23,24} However, such models fall short in reproducing the optical response properties of realistic plasmonic substrates.^{19,25} To overcome this limitation, multiscale approaches have been developed that couple a QM description of the molecule with classical electromagnetic models of the plasmonic environment. Continuum models, such as the Boundary Element Method (BEM),²⁶ which represent the nanostructures as a continuum dielectric, have been adopted for this purpose.^{19,27–32} Nevertheless, to fully account for atomistic details such as edges, tips, defects, and junctions, which are critical to achieving high local

Received: November 4, 2025
Revised: December 19, 2025
Accepted: January 6, 2026
Published: January 22, 2026



field enhancement, atomistic methodologies must be exploited.^{33–40} In this framework, we have recently introduced a multiscale QM/classical approach for simulating SERS spectra,³⁹ based on the fully atomistic frequency-dependent fluctuating charges (ω FQ)^{41,42} and dipoles (ω FQF μ)⁴³ approaches. These models, rooted in the Drude conduction theory with phenomenological corrections for quantum tunneling, enable the simulation of the optical response of a wide range of plasmonic materials, including alkali metals,⁴¹ noble metal nanoparticles,^{43–45} nanoalloys,⁴⁴ and graphene-based substrates,⁴² even in the presence of structural defects,^{46,47} solvent effects,⁴⁵ and subnanometer gaps.^{41,43,48} Their classical formulation allows simulations of systems with more than one million atoms,⁴⁹ yielding an almost perfect agreement with full *ab initio* methods,^{37,41,43} while retaining a fully atomistic resolution.

In this work, we extend the QM/ ω FQ(F μ) approach to SEIRA spectroscopy and apply it to adenine, one of the DNA nucleobases, adsorbed on gold nanostructures and graphene disks. Adenine represents a particularly relevant test system for SEIRA studies, as it has been extensively investigated both experimentally and theoretically.^{14,50–53} SEIRA spectra of adenine have been reported on a variety of gold-based substrates, including Au(111) and Au(100) electrodes,^{8,50,54,55} as well as silica core–gold nanoshells.⁵⁶ More recently, SEIRA on graphene oxide (GOEIRA) has also been explored, highlighting the potential of carbon-based platforms.⁵⁷

The manuscript is organized as follows. First, the computational framework used to investigate SEIRA and SERS spectra is introduced. The approach is then applied to predict the SEIRA signals of adenine (ADE) nucleobase in proximity to gold nanostructures, with a focus on the influence of molecular orientation and adsorption site on the spectral response. The results are compared with SERS spectra and validated against the available experimental data. In the final section, we assess the potential of graphene as a SEIRA substrate by analyzing how its structural and electronic properties modulate the molecule-graphene response in the IR range. Summary, conclusions, and future perspectives end the paper.

2. METHODS

SEIRA spectra are computed using a multiscale QM/classical approach. In this framework, the plasmonic nanostructure and its interaction with external radiation, leading to the formation of localized surface plasmons, are described through the fully atomistic ω FQ and ω FQF μ models, whereas the molecular subsystem is treated quantum mechanically at the DFT and TDDFT levels. In this section, we briefly recall the theoretical foundations of ω FQ and ω FQF μ , and then present their coupling to a QM region and its extension to SEIRA simulations.

2.1. ω FQ and ω FQF μ Models for Nanoplasmonics

ω FQ is a classical, fully atomistic, and frequency-dependent model able to describe the plasmonic features of various materials, including alkali metals⁴¹ and graphene-based materials.⁴² In ω FQ, each atom is endowed with a complex electric charge, and the charge exchange between the atoms is assumed to be governed by a Drude-conduction mechanism mediated by a phenomenological damping, which limits the charge flow between nearest neighbors and mimics the physics of quantum tunneling.⁴¹ The charges are obtained by solving the following linear system defined in the frequency domain⁴⁹

$$[\bar{\mathbf{K}}\mathbf{T}^{\text{qq}} - z_q(\omega)\mathbf{I}_N]\mathbf{q}(\omega) = -\bar{\mathbf{K}}\mathbf{V}^{\text{ext}}(\omega) \quad (1)$$

\mathbf{I}_N is the identity matrix of order N , where N is the number of atoms. $\bar{\mathbf{K}}$ is a $N \times N$ matrix whose elements read

$$\bar{K}_{ij} = K_{ij} - \sum_k K_{ik}\delta_{kj} \quad (2)$$

where δ_{ij} is the Kronecker delta. The elements of the \mathbf{K} matrix take the following form

$$K_{ij} = \begin{cases} \frac{[1 - f(r_{ij})]\mathcal{A}_i}{r_{ij}} & \text{if } i \neq j \\ 0 & \text{if } i = j \end{cases} \quad (3)$$

The left-hand side of eq 1 is composed of a real frequency-independent matrix with a diagonal shift of the complex scalar frequency-dependent $z_q(\omega)$, which is a function of the relaxation time τ , a friction-like constant that accounts for scattering events, and the electron density of the system, n

$$z_q(\omega) = -\frac{\omega}{2n\tau}(\omega\tau + i) \quad (4)$$

The charge flow between atoms results from the Drude conduction mechanism, the interaction between charges, represented by the \mathbf{T}^{qq} interaction kernel,⁴¹ and with the external potential \mathbf{V}^{ext} associated with the external electric field oscillating at frequency ω . K_{ij} is a real symmetric Drude matrix, expressed in terms of the effective area \mathcal{A}_i of atom i . $f(r_{ij})$ is a Fermi-like function effectively modeling quantum tunneling effects

$$f(r_{ij}) = \frac{1}{1 + \exp\left[-d\left(\frac{r_{ij}}{s r_{ij}^0} - 1\right)\right]} \quad (5)$$

where r_{ij} is the distance between atoms i and j .⁴¹

ω FQ can be extended to treat graphene-based nanomaterials,⁴² by introducing the effective mass m^* (which is 1 au for pure metals).⁵⁸ The 3D atomic effective electron density of graphene n reads

$$n = \frac{\tilde{n}_0}{m^*} = \sqrt{\frac{n_{2D}}{\pi}} a_0 v_F = \frac{E_F a_0}{\hbar\pi} \quad (6)$$

where \tilde{n}_0 is the 3D atomic electron density, v_F is the Fermi velocity, E_F is the Fermi energy, a_0 is the Bohr radius, and n_{2D} is the two-dimensional electron density of graphene.^{42,58} In eq 6, we have used the relationship between E_F and n_{2D} : $E_F = \hbar v_F \sqrt{\pi n_{2D}}$.^{42,58} eq 6 shows that, for graphene-based nanostructures, the Drude dynamics also depends on the Fermi energy, which can be tuned experimentally.^{15,42,46,47,59–62}

While ω FQ can properly describe alkali metals and graphene, it fails to account for the optical response of noble metals, as it assumes that the conduction electrons follow a purely Drude-like behavior and neglects interband transitions.^{63–65} To overcome this limitation, we introduce an additional polarization source that accounts for the polarizability of the d -shell.⁶⁶ In the resulting ω FQF μ approach,⁴³ each atom is represented by a charge q and a dipole μ , the latter depending on the interband frequency-dependent atomic polarizability $\alpha_{\text{IB}}(\omega)$. The equations of motion of charges and dipoles are coupled by solving the following linear system of equations

$$\begin{pmatrix} \bar{\mathbf{K}} & \mathbf{0} \\ \mathbf{0} & \mathbf{I}_{3N} \end{pmatrix} \begin{pmatrix} \mathbf{T}^{\text{qq}} & \mathbf{T}^{\text{q}\mu} \\ \mathbf{T}^{\mu\text{q}} & \mathbf{T}^{\mu\mu} \end{pmatrix} - \begin{pmatrix} z_q(\omega)\mathbf{I}_N & \mathbf{0} \\ \mathbf{0} & z_\mu(\omega)\mathbf{I}_{3N} \end{pmatrix} \begin{pmatrix} \mathbf{q}(\omega) \\ \boldsymbol{\mu}(\omega) \end{pmatrix} = \begin{pmatrix} \bar{\mathbf{K}} & \mathbf{0} \\ \mathbf{0} & \mathbf{I}_{3N} \end{pmatrix} \begin{pmatrix} -\mathbf{V}^{\text{ext}}(\omega) \\ \mathbf{E}^{\text{ext}}(\omega) \end{pmatrix} \quad (7)$$

where

$$z_\mu(\omega) = -\frac{1}{\alpha_{\text{IB}}(\omega)} \quad (8)$$

The left-hand side of the linear system in eq 7 comprises all of the interaction kernels: charge–charge \mathbf{T}^{qq} , charge–dipole $\mathbf{T}^{\text{q}\mu}$, dipole–charge $\mathbf{T}^{\mu\text{q}}$, and dipole–dipole $\mathbf{T}^{\mu\mu}$.^{67,68} The right-hand side of the

system contains the external sources, i.e., the potential (V^{ext}) and the electric field (E^{ext}).

2.2. QM/ ω FQF μ for SEIRA

QM/ ω FQF μ has recently been extended to the damped linear response formalism to compute complex polarizabilities, from which SERS signals can be calculated.³⁹ All of the technical details on the derivation of the formalism in the time-dependent Kohn–Sham (TDKS) framework, as well as the details of the ground-state (GS) coupling, can be found in ref 39.

The QM/ ω FQF μ approach for SERS spectra assumes that the perturbation operator includes both the electric potential of the external field and the local field operator accounting for the plasmon-induced field generated by the substrate.^{28,34} These effects are incorporated into the effective KS operator through the image field term, describing the polarization induced by the nanostructure's charges and dipoles, and the local field term due to the plasmon, both evaluated self-consistently with the perturbed TDKS density.^{28,34} Solving the coupled-perturbed TDKS equations yields the frequency-dependent complex molecular polarizability tensor, which naturally includes the scattered and reflected fields responsible for the electromagnetic enhancement observed in SERS. A similar physical picture is followed here to extend the QM/ ω FQF μ to SEIRA intensities. In particular, in line with ref 69, IR absorption intensities can be calculated from the geometrical derivatives of an “external” dipole moment, which is the sum of the gas-phase molecular dipole moment, \vec{d} , and an additional dipole moment, \vec{d} , induced on the nanostructure by the molecular density, i.e., that accounts for local field effects.^{70,71} The induced dipole \vec{d} can be expressed as

$$-\vec{d} \cdot E^{\text{ext}} = \sum_{p=1}^N [q_p V^{\text{ext}}(\mathbf{r}_p) - \mu_p \cdot E^{\text{ext}}(\mathbf{r}_p)] \quad (9)$$

where q_p and μ_p , located at position \mathbf{r}_p , represent charges and dipoles induced by the GS QM density on the plasmonic substrate. These are calculated through eq 7 by substituting the external field with the electric potential V^{QM} and field E^{QM} generated by the oscillating molecular density. Thus, the induced dipole is obtained from the GS QM electric potential and field, and the local field charges and dipoles oscillate at the normal-mode frequency.

2.3. Computational Details

QM/ ω FQ(F μ) is applied to the calculation of SERS and SEIRA spectra of the adenine (ADE) nucleobase adsorbed on gold and graphene substrates. ADE is chosen because it is of broad interest in biosensing due to its role as a DNA building block.^{14,72–75} The gold nanoparticle is characterized by an icosahedral morphology, comprising 10,179 atoms (Au₁₀₁₇₉ Ih) with a radius of about 3.83 nm and 14 gold shells, and its dipolar plasmon resonance frequency (PRF) is 2.21 eV (calculated at the ω FQF μ level). Two additional gold substrates, Au₄₉₀₄₉ Ih and Au₁₀₄₂₂₃ Ih, are considered, with PRFs falling at 2.21 and 2.18 eV, respectively, to showcase the dependence of SERS and SEIRA spectra by enlarging the nanostructure. For graphene substrates, we consider graphene disks, with geometries taken from ref 46. and³⁹, characterized by diameters ranging from 24 nm (17,269 atoms) to 100 nm (300,695 atoms) and a Fermi energy of 0.40 eV. In addition, we examine a 32 nm graphene disk for which the Fermi energy is tuned from 0.40 to 0.09 eV. The gold and graphene substrates are treated at the ω FQF μ and ω FQ levels by exploiting the parameters from refs 43 and 42, respectively.

In all molecule–substrate configurations, the 9H tautomer of adenine (ADE), shown in Figure 1, is described at the DFT level by using the BP86 functional and the TZP basis set, in agreement with previous studies,^{34,39} with the default DFT integration grid for adenine on gold and a finer grid for adenine on graphene (see also Figure S1a in Supporting Information – SI). The distance between the ADE nitrogen atom and the closest gold atom of the NP is set to 3 Å. For specific configurations, v_{N9} , v_{N10} , f_{N1} , f_{N7} , and f_{N10} , the N–Au distance is increased to 3.4 Å to avoid steric interactions between molecular Hydrogen and Au atoms.

Normal-mode displacements are obtained from the isolated QM molecule since the effect of the plasmonic substrate is negligible.^{34,39}

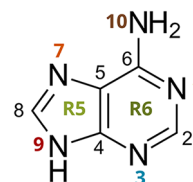


Figure 1. Molecular structure and atom labeling of adenine.

Geometrical derivatives of frequency-dependent dipoles (SEIRA) and polarizabilities (SERS) are obtained using a three-point central numerical procedure with a differentiation step of 0.001 Bohr. For completeness, Figure S1b in the SI reports a numerical stability test carried out with a step size of 0.0005 Bohr. SERS spectra are simulated by setting the Raman incident field at the PRF of the substrate, and we use a lifetime of 0.10 eV, in agreement with previous studies.^{34,39} SERS and SEIRA raw data are convoluted by using a Lorentzian band shape with a full width at half-maximum (fwhm) of 10 cm⁻¹. All QM/ ω FQ and QM/ ω FQF μ calculations are performed by using a locally modified version of the AMS software.^{76,77} By following the procedure reported in ref 39, for SERS, QM/FQ(F μ) contributions to the GS were neglected, as their effect on SEIRA signals is only marginal (see also Figure S2 in the SI).

To quantify spectral enhancements due to the presence of the plasmonic nanostructure,³⁹ we introduce the averaged enhancement factor (AEF) and the maximum enhancement factor (MEF), which are computed from the IR/Raman enhancement factor (EF) associated with each i -th normal mode

$$EF^i(\omega) = \frac{I^i(\omega)}{I_{\text{vac}}^i(\omega)}; \text{AEF}(\omega) = \frac{\sum_i I^i(\omega)}{\sum_i I_{\text{vac}}^i(\omega)}; \text{MEF}(\omega) = \max_i EF^i(\omega) \quad (10)$$

The i -th normal mode showing the maximum enhancement factor is indicated as i -MEF.

3. RESULTS AND DISCUSSION

3.1. SEIRA and SERS Spectra of Adenine on Gold Nanostructures

Multidentate binding modes render the preferred orientation of ADE on gold and silver surfaces challenging to resolve.⁵² In fact, in its 9H tautomer form, adenine has five nitrogen atoms as potential coordinating sites for metal surfaces. Several adsorption models for adenine have been proposed in the literature.⁵² Adsorption via the N3/N9 side (see Figure 1 for atom labeling) is supported by combined SERS and SEIRA spectra on gold nanoshells,¹⁴ electrochemical tip-enhanced Raman spectroscopy (EC-TERS) studies on protonated adenine,⁷⁸ X-ray absorption and DFT calculations,⁵¹ and SERS spectra on gold nanoparticles.⁷⁹ Adsorption via N7/N10 sites is backed by SEIRA and cyclic voltammetry studies on gold electrodes,^{50,54,55} where surface-enhanced ring stretching modes (N7–C5, N7–C8) and scissoring of the amino group were observed. Similarly, the N1/N10 adsorption model is supported by subtractively normalized interfacial Fourier transform infrared spectra (SNIFTIRS) on Au(111)⁸⁰ and EC-TERS studies for deprotonated adenine at higher potentials, suggesting a tilted flat orientation.⁷⁸ Although recent SERS and SEIRA experimental evidence proposes a vertical or tilted orientation of adenine on gold,⁷⁹ computational studies, relying on molecular dynamics simulations, indicate that a flat orientation may occur at low concentrations.^{81,82} To account for all possible adsorption sites, in this work, we consider six binding modes of ADE on gold: four “end-on” modes involving

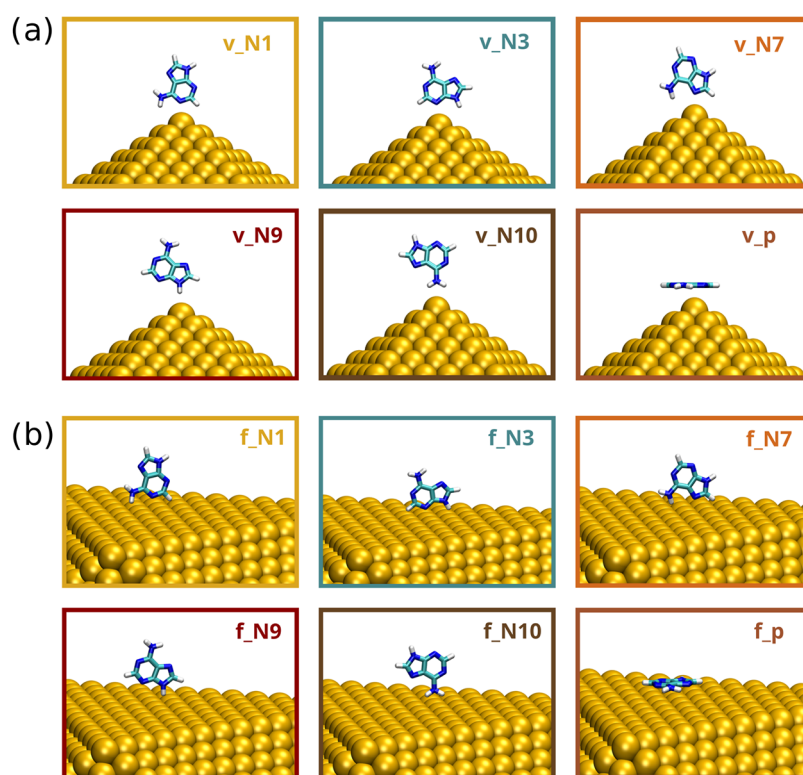


Figure 2. Molecular configurations of ADE adsorbed on the vertex (a) and face (b) positions of the Au₁₀₁₇₉ Ih NP.

each purine ring nitrogen atoms (N1, N3, N7, and the protonated N9), one “face-on” mode with the adenine rings oriented parallel to the surface (p), and one mode involving the exocyclic amino group (N10).¹⁴ To investigate the influence of these binding orientations at different adsorption sites on a model Au₁₀₁₇₉ Ih nanostructure, 12 configurations are generated, six at the vertex and six at the face of the gold nanostructure, as shown in Figure 2a,b. The corresponding structures are named using the following nomenclature: v_{atom} (vertex) or f_{atom} (face), where *atom* indicates ADE atom binding to the metal surface (see Figure 1 for atom labeling). We then exploit QM/ ω FGF μ to analyze the sensitivity of SEIRA and SERS signals to the specific adsorption sites of various ADE orientations on the Au NP. The resulting spectra are graphically reported in Figure 3, where they are compared with IR and Raman spectra of the isolated molecule, which are used as a reference to quantify the surface-induced variations.

In the gas phase, the ADE IR spectrum (black dashed line in Figure 3a) features four main peaks: the most intense (1605 cm^{-1}) corresponds to NH₂ scissoring and C5–C6 and C6–N10 stretching modes, and is accompanied by a shoulder peak at 1575 cm^{-1} , which is associated with N9–H and C8–H bending and N3–C4, N1–C6, C5–N7, and N7–C8 stretching (see Figure S3 in the SI for a graphical depiction of ADE normal modes). Less intense bands are present at 1448 cm^{-1} , assigned to the stretching of N7–C8, N1–C6, and C2–N3 and the bending of C2–H and C8–H, and at 1292 cm^{-1} , corresponding to the stretching of C2–N3 and C5–N7, together with the bending of C2–H, C8–H, and N9–H.

Most SEIRA spectra closely resemble the gas-phase IR spectrum, especially v_{N3} , v_{N9} , and v_{N10} (vertex configurations) and f_{N3} , f_{N9} , and f_{N10} (face configurations), suggesting that the IR-active vibrational modes of ADE are largely preserved upon adsorption. All configurations

exhibit their most intense peak at 1605 cm^{-1} , except for v_{N1} and f_{p} configurations. In particular, v_{N1} presents the dominant peak at 1292 cm^{-1} , which corresponds to an in-plane normal mode involving stretching of the C2–N3 and C5–N7 bonds, along with bending motions of the C2–H, C8–H, and N9–H bonds. In contrast, for f_{p} , the strongest signals correspond to out-of-plane modes at 808 cm^{-1} (C8–H wagging) and 932 cm^{-1} (C2–H wagging), the latter also reporting the MEF (5.9, see Table S1 in the SI). Such bands are also particularly intense for the v_{p} configuration. The f_{N1} configuration also deviates from the typical pattern, displaying two intense peaks at 1292 and 1575 cm^{-1} , linked to pronounced deformations of the adenine rings. Remarkably, the spectra obtained for ADE adsorbed on the vertex or face binding the same atom to the metal surface display a similar spectral profile, suggesting the relevant role of the binding site. To further deepen this point, in Figure S4 in the SI, we report the SEIRA spectrum as obtained by simply averaging the spectra over the 6 configurations on the vertex and the face of Au₁₀₁₇₉ Ih. The two obtained spectra are characterized by the same profile, confirming that the adsorption site (face or vertex) does not significantly alter the overall spectral shape.

The face and vertex adsorption configurations are associated with substantial differences in the enhancement factors. In fact, as can be appreciated by Figure 3a and Table S1 in the SI, the MEFs calculated for ADE adsorbed on the vertex of the gold icosahedron are generally 1 order of magnitude larger than the corresponding values computed for face dispositions. This is not surprising and is related to the so-called tip effect, which arises from the highly localized and inhomogeneous electric fields at metallic nanostructures “hot spots” (sharp tips and vertices).^{1,83} The normal modes associated with MEF (*i*-MEF) in most configurations are in-plane vibrations (see Table S1 in the SI), which is expected considering that the NP induced electric field

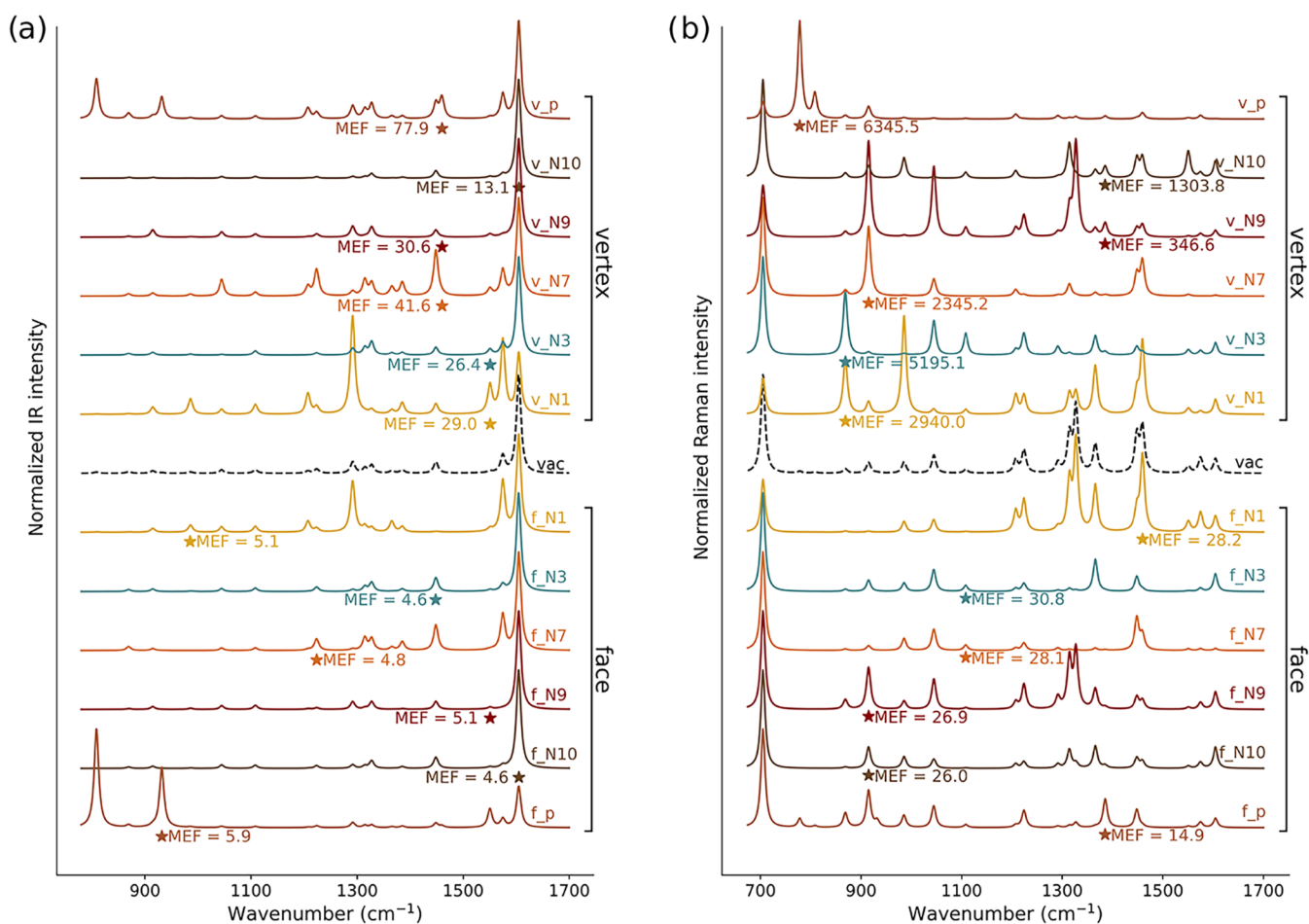


Figure 3. Normalized $QM/\omega FQF\mu$ SEIRA (a) and SERS (b) spectra of the six configurations of ADE adsorbed on the vertex and face of Au_{10179} Ih. SERS spectra are calculated at the PRF of the gold substrate (2.21 eV/560 nm). Stars denote the *i*-MEF. MEF values are also reported.

is supposed to display the largest variations perpendicularly to the gold surface. It is also worth noting that the most intense band of the spectrum is generally not associated with the normal mode, reporting the largest MEF, except for v_N10 (MEF = 13.1) and f_N10 (MEF = 4.6).

An enhancement of the IR signal (AEF) is observed across all configurations except for v_p , where the IR signal is overall quenched (AEF < 1, see Table S1 in the SI). For the latter structure, a strong enhancement is detected at 1460 cm^{-1} (involving the NH_2 group) which also dominates the MEF profile in v_N7 and v_N9 (see Figure S5 in the SI). This suggests a particularly favorable orientation of the NH_2 oscillating dipole relative to the local electromagnetic field. Notably, the MEF of the 1460 cm^{-1} band in these three configurations is the largest across all of the considered geometries. The raw data reported in Table S1 in the SI also highlight that SEIRA spectra for face configurations show lower AEFs compared to vertex configurations, coherently with the comment above regarding the MEFs. Interestingly, by plotting the enhancement factors for each normal mode (see Figure S5b in the SI), we note that, for the face configurations, the enhancement factors are uniform across vibrational modes, clustering around the AEF, with the *i*-MEFs depending on the specific binding pose of the molecule. On the contrary, the vertex configurations plots (Figure S5a in the SI) are dominated by the specific normal modes that report the MEF. This is again a consequence of the so-called “tip effect.” Overall, our

calculated AEFs average around 5, with v_N10 showing the highest value (9.2), and MEFs reach approximately 30, which aligns well with the typical SEIRA enhancement range, generally expected on the order of 10–100.^{17,18,20}

We now move on to comment on the SERS signals of ADE adsorbed on Au NP. The gas phase Raman spectrum (see Figure 3b) is characterized by a dominant peak at 705 cm^{-1} , corresponding to the “breathing” of the purine rings, along with characteristic peaks in the $1300\text{--}1500\text{ cm}^{-1}$ region,⁸⁴ primarily in-plane bending normal modes, most of which involve the NH_2 group. In Figure 3b, the SERS spectra of each adsorption configuration are graphically reported. Each vertex geometry displays a distinct most intense peak: 986 cm^{-1} for v_N1 , 705 cm^{-1} for v_N3 , v_N7 , and v_N10 , 1327 cm^{-1} for v_N9 , and 778 cm^{-1} for v_p . Differently, most face configurations report the most intense peak at 705 cm^{-1} (ring breathing), with the exception of f_N1 . The f_N1 SERS spectrum is in fact dominated by peaks in the region $1200\text{--}1500\text{ cm}^{-1}$, with the most intense peak appearing at 1327 cm^{-1} (in-plane stretching mode involving C8–N9, C6–N1, and N3–C4 bonds, along with bending of C8–H and N9–H). This highlights a significant difference with respect to SEIRA (Figure 3a). In fact, in SEIRA, the spectral profile is mainly determined by the binding atom. The same is not valid for SERS, for which face and vertex spectra of configurations exploiting the same binding site substantially differ. Furthermore, the SERS spectra of face structures mostly resemble the spectral profile of ADE in

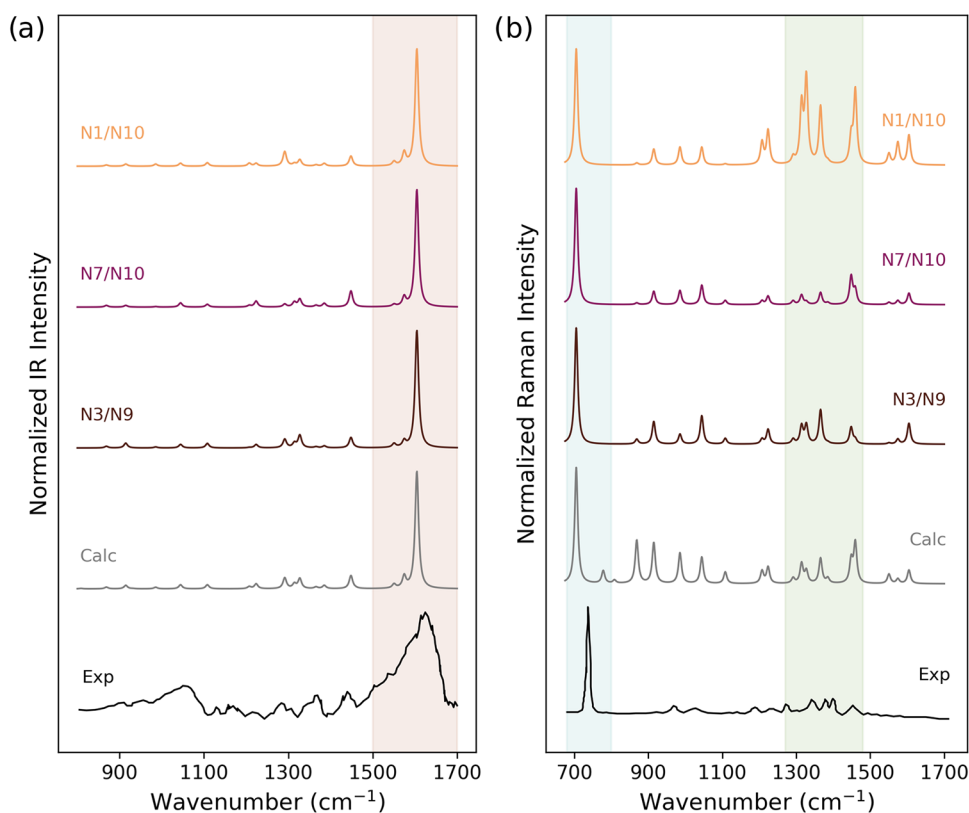


Figure 4. (a) Normalized QM/ ω FQF μ SEIRA and (b) SERS spectra, calculated by averaging N3/N9, N7/N10, N1/N10, and all (Calc) configurations of ADE on Au₁₀₁₇₉ Ih, together with the experimental results (Reproduced from ref 14. Copyright 2009 American Chemical Society.). The characteristic regions of SEIRA and SERS spectra are highlighted.

vacuo. Figure S6b in the SI further illustrates this trend by plotting the enhancement factors as a function of the normal mode. For face configurations, these are relatively uniform across Raman bands, reflecting a similar behavior as in SEIRA. On the contrary, the SERS spectra of vertex structures substantially deviate from the gas-phase, also activating quasi-inactive normal modes in the gas phase. As an example, we highlight the peak at 778 cm⁻¹ (out-of-plane ring deformation) for the v_p configuration for which the highest MEF (6345) is reported. This can also be appreciated in Figure S6b in the SI, where the vertex configurations display a selective enhancement profile as a function of the normal mode, with EF values varying by up to 2 orders of magnitude. To further highlight the effects of the adsorption site, as for SEIRA, the SERS spectra averaged on the 6 face and 6 vertex configurations, separately, are reported in Figure S4 in the SI. Different from SEIRA, the two averaged spectra show significant discrepancies, highlighting, in this case, the substantial influence of the nanostructure-molecule morphology and the adsorption site on the spectral profile.

The differences between face and vertex configurations are also reflected in the calculated values of AEF and MEF. As for SEIRA, the SERS enhancement factors are generally lower when ADE is adsorbed on the face than on the vertex. In fact, AEF and MEF values for vertex configurations are, respectively, one and two orders of magnitude higher than those observed for face configurations. In most configurations, the MEF is associated with normal modes involving in-plane atom displacements, as expected, considering the electric field gradient induced by the plasmon excitation in the nanostructure. As mentioned above, the only exception is v_p, which is characterized by a flat adsorption on the Au NP vertex, thus enhancing mostly out-of-

plane normal modes, although providing an overall small AEF (47.6). The average AEF and MEF values are one and 2 orders of magnitude higher than those observed for SEIRA, and are of the order of 10² and 10³ (vertex) and 10¹ and 10² (face). This is consistent with the electromagnetic enhancement theory, for which in SERS, the local electromagnetic enhancement scales with the fourth power of the local induced field, whereas in SEIRA, it scales with the square.

We finally note that the absolute values of AEF and MEF of vibrational spectroscopies depend on the nanoparticle size because of the larger induced field in the proximity of the nanostructure surface.^{43,85,86} To showcase such a dependence, in Figure S8 in the SI, we report the SEIRA and SERS spectra of ADE in v_{N7} configuration on Au nanostructures composed of 49,049 (radius = 6.57 nm) and 104223 (radius = 8.49 nm) (see Figure S7 in SI). By increasing the NP radius, the AEF and MEF increase by approximately a factor of 2 in SEIRA and 4.5 in SERS; however, all SEIRA and SERS spectra maintain the same profile as that computed for the smallest nanoparticle, demonstrating the robustness of our approach.

Figure 3 demonstrates that SEIRA and SERS selectively amplify different vibrational bands, highlighting the complementarity of the two techniques, which has been widely exploited to understand the orientation of adsorbed adenine on metal substrates.⁵² As commented above, three main adsorption models, which depend on the specific binding site of ADE on the Au surface, have been proposed in the literature and validated through various spectroscopic techniques:^{50,51,54–56,78–80} N3/N9, N7/N10, and N1/N10.

To shed light on the most favorable configuration, in Figure 4, we average the computed SEIRA and SERS spectra obtained

from the corresponding sets of adenine-gold configurations (N1/N10; N7/N10; N3/N9), together with the total averaged computed spectra (Calc), and the experimental data (Exp) of adenine adsorbed on gold nanoshells at neutral pH reported in ref 14. The experimental SEIRA spectrum (Figure 4a, bottom) is dominated by the in-plane symmetric NH_2 scissoring mode (1625 cm^{-1}), and the in-plane ring modes ($1058, 1286, 1310, 1595\text{ cm}^{-1}$), which is generally explained by suggesting that the C6- NH_2 group is oriented perpendicularly to the surface with the amino group far from the surface.¹⁴ The experimental SERS spectrum (Figure 4b, bottom) shows a prominent “ring breathing” mode at 735 cm^{-1} , but it is also characterized by the presence of a spectral fingerprint in the region $1300\text{--}1500\text{ cm}^{-1}$ (in particular peaks 1307 and 1337 cm^{-1}), and the presence of the weak out-of-plane ring mode at 787 cm^{-1} suggests a slightly tilted adenine orientation.¹⁴

The spectra resulting from the three adsorption models show a slight variation in the relative intensity of the peaks, in particular, around 1300 cm^{-1} for SEIRA and within the $1200\text{--}1500\text{ cm}^{-1}$ range for SERS, which are associated with the differences in the enhancement profiles discussed above. Based on our results, the N1/N10 configuration appears to be the least accurate in reproducing the experimental findings, as it fails to reproduce the experimental features. By contrast, both the N7/N10 and N3/N9 models provide a satisfactory match with experiment: in particular, the N3/N9 configuration offers the best agreement with the experimental SERS spectrum, successfully reproducing both the “ring breathing” band at 735 cm^{-1} and the structured features in the $1300\text{--}1500\text{ cm}^{-1}$ region, and its SEIRA spectrum also gives good agreement with the experimental profile in terms of relative intensity. Our analysis thus suggests that N3/N9 or N7/N10 are the most likely adsorption models. Finally, in order to also consider all of the configurations that can possibly contribute to the experimental spectrum, we compare the total average spectra, denoted as “Calc,” with the experiment (Figure 4). This remarkably captures most experimental features with a good reproduction of the overall spectrum, although some discrepancies are reported, such as the overestimated intensity in the $900\text{--}1100\text{ cm}^{-1}$ region of the calculated SERS spectrum.

3.2. SEIRA Spectra of Adenine on Graphene

Recently, IR spectra of small aromatic molecules, such as rhodamines and purines, have been recorded on pristine graphene,⁸⁷ graphene oxide,⁵⁷ and carbon dots,⁵³ reflecting the growing interest in graphene and its derivatives as potential SEIRA substrates.^{2,3,88} This interest arises from the possibility of tuning graphene plasmons to fall in the mid-IR region, i.e., in the region of molecular vibrations, consequently enhancing the corresponding absorption signals.¹⁵ To achieve resonance enhancement in the infrared range, the PRF of graphene substrates can be tuned by either changing their size or geometry or adjusting their carrier density through electrical gating or chemical doping, i.e., by varying the Fermi energy E_F .

To showcase the capability of QM/ ω FQ to calculate SEIRA spectra, we adsorb ADE on disk-shaped graphene nanostructures with diameters ranging from 24 to 100 nm (GD24–GD100). An adsorption geometry with ADE placed parallel to the graphene disk, at a distance of 3.5 \AA , is considered (see Figure 5), following experimental and computational evidence.^{89–92} For comparison, a geometry in which the N1 atom is oriented toward the graphene disk, at a distance of 3.5 \AA , is also investigated; the corresponding results are reported in the

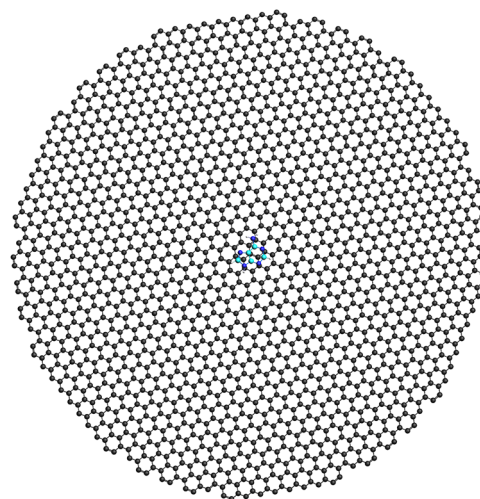


Figure 5. Graphical depiction of ADE adsorbed on a graphene disk in a parallel configuration.

SI (Figures S8, S9, and S11). In Figure 6, we first focus on the dependence of SEIRA signals on the structural parameters of the

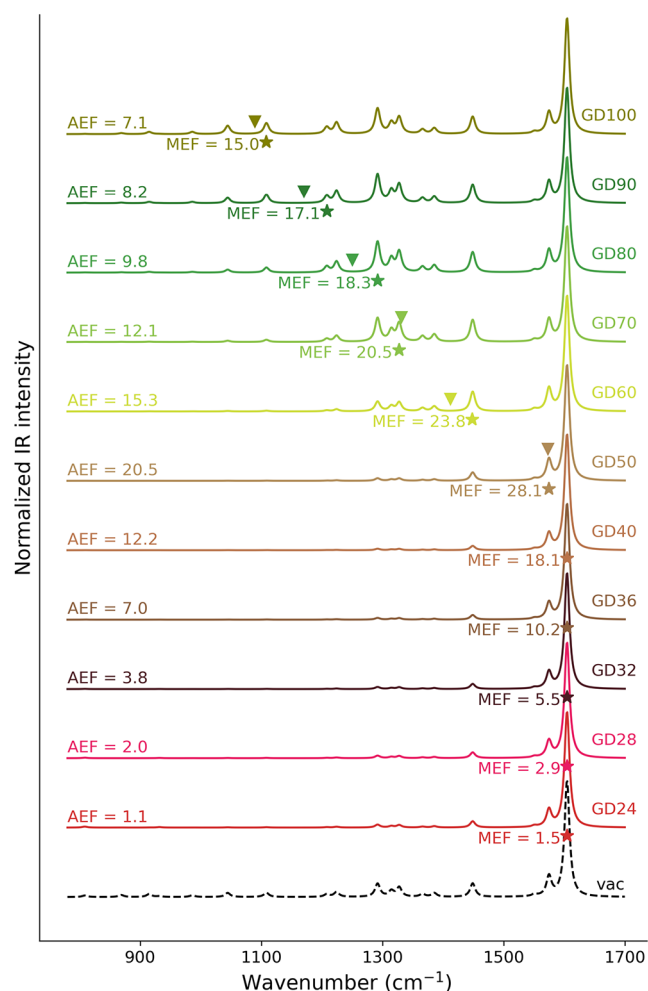


Figure 6. Normalized QM/ ω FQ SEIRA spectra of ADE in the parallel configuration adsorbed on GDs of increasing size. Stars denote the i -MEFs; triangles indicate the PRFs. AEF and MEF values are also reported.

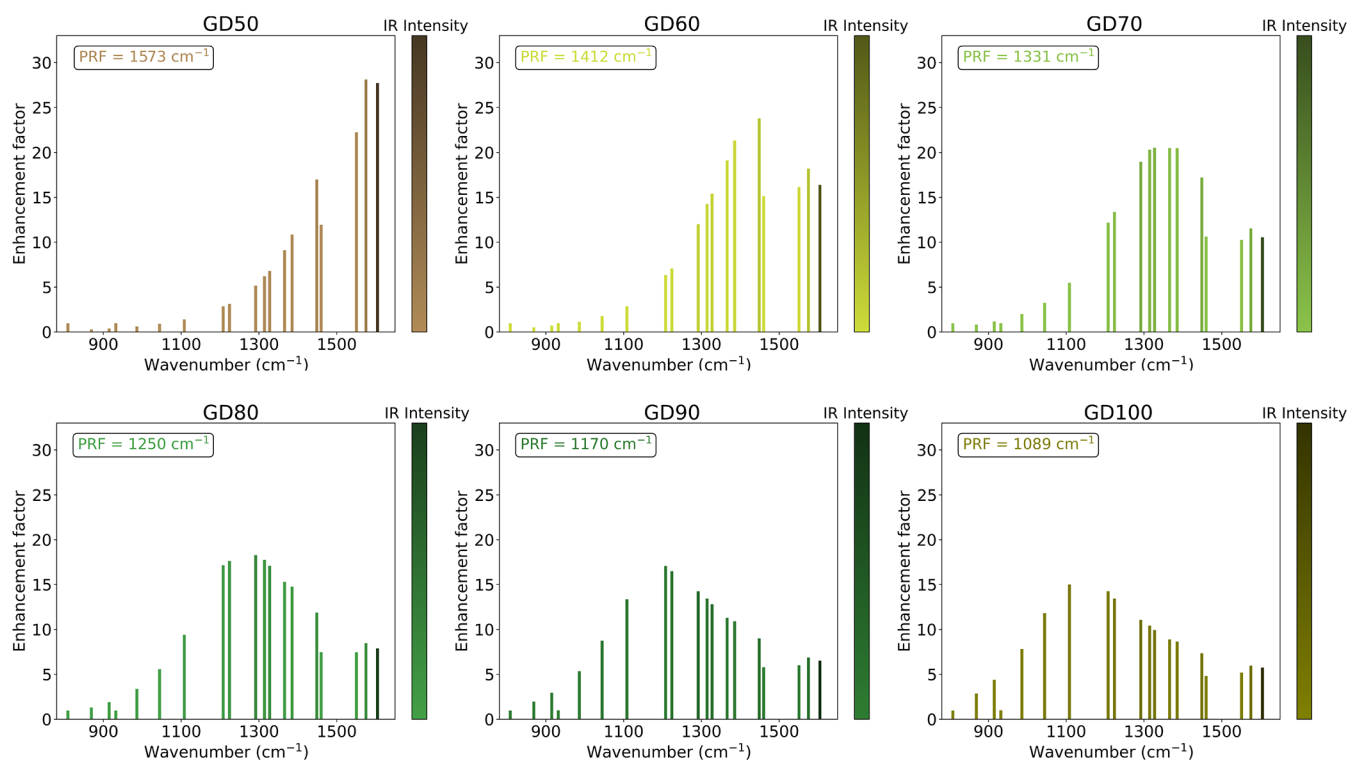


Figure 7. SEIRA enhancement factors computed for each normal mode of ADE in the parallel configuration on GDs of increasing size. EFs are plotted with a palette following the SEIRA intensities.

graphene substrate (see Table S2 in the SI for the PRF of the various structures). The Fermi energy of the graphene disks is set to 0.4 eV, in line with previous studies.^{39,42,46}

The most intense peak for all GD diameters appears at 1605 cm^{-1} , which corresponds to the NH_2 scissoring and six-membered-ring stretching modes, similarly to the gas-phase spectrum (black dashed line). The SEIRA spectra show a clear dependence on GD size. For the smallest GDs (diameter <40 nm), the IR peaks below 1400 cm^{-1} are characterized by a very low intensity with respect to the gas-phase spectrum, indicating a partial deactivation of ADE's normal modes at lower wavenumbers. As the disk size increases (for diameter >60 nm), these peaks gradually become more intense, and the spectra of larger disks become increasingly similar to the gas-phase IR spectrum, with its main spectral features largely recovered, even if with different relative intensities. This trend, which is also observed for the N1 configuration (see Figure S8 in SI), can be rationalized by considering that the most enhanced peaks shift to lower energy as the disk diameter increases, directly following the trend observed for the PRF (see Tables S2 and S4 in SI). In fact, for structures with a diameter <40 nm, which report the PRF in the range 1734–2258 cm^{-1} , the most intense peak at 1605 cm^{-1} also coincides with the *i*-MEF. From GD50 onward, the *i*-MEF shifts to normal modes that resonate with the PRF of the graphene disk, which progressively redshifts with increasing disk size.⁴² Such a distribution can be better appreciated by plotting the enhancement factors for each normal mode as a function of the disk size (from GD50 to GD100), which is reported in Figure 7. For larger disks, where the PRF aligns with the considered IR spectral range of adenine, the most enhanced normal modes shift to remain in resonance with the PRF of the respective disk. For example, in GD60, the PRF is 1412 cm^{-1} , and the dominant enhancements occur near 1400 cm^{-1} , with *i*-MEF corresponding to the peak at 1448 cm^{-1} . A similar GD size-

driven behavior is observed for ADE in perpendicular N1 configuration (see Figure S9 in SI); however, for the largest disks (GD80, GD90, and GD100), the *i*-MEF no longer shifts with disk size and consistently corresponds to the same in-plane bending mode at 1224 cm^{-1} .

Figure 6 also reports the average and maximum enhancement factors computed for each structure (see also Figure S7 and Table S4 in the SI). For all of the structures, we report an enhancement of the SEIRA signals (AEF > 1.0). In particular, the largest AEF (20.5) and MEF (28.1) are reported for GD50. For larger disks than GD50, both AEF and MEF values decrease monotonically (see also Figure S10 in the SI). By looking at Figure S11 in SI, we notice that, for ADE in the perpendicular geometry, AEF and MEF trends are slightly altered: AEF and MEF reach their maximum values for GD50 (15.05) and GD60 (23.64), respectively, and then decrease with an increasing disk size. These trends can be rationalized by considering that as the disk radius increases, the induced plasmon field experienced by ADE at the center of the graphene disk becomes more homogeneous. At the same time, the edges, where the electric field is maximally enhanced, are farther away from the ADE molecule with a larger disk radius. Such effects counterbalance the resonance conditions created by enlarging the size of the disk (shifting the PRF to resonate with ADE vibrational modes). This is indeed in line with what has been previously commented on for SERS on graphene disks.³⁹

As a final comparison, we note that available experimental data from ref 57 report that the IR spectrum of adenine on graphene oxide exhibits an AEF of approximately 30. In particular, the MEF (36.9) is attributed to C=N stretching at 1418 cm^{-1} , while other enhanced peaks correspond to C–H bending modes (1241 and 1394 cm^{-1}), C–N stretching (1080 cm^{-1}), and NH_2 scissoring and aromatic rings stretching mode (1604 cm^{-1}).⁵⁷ Although we focused on an ideal graphene

substrate, our results are indeed in good agreement with the experimental values.

To conclude this section, we study the SEIRA dependence of ADE in the parallel configuration adsorbed on a graphene disk with a fixed diameter (32 nm) as a function of E_F . In particular, we exploit a peculiar property of graphene substrate: by reducing the electron density (i.e., by lowering the Fermi energy), the PRF redshifts.^{42,47} This allows us to mimic the electronic and optical properties of large nanodisks by using smaller nanostructures since such a feature is equivalent to increasing the size of the graphene disk. Specifically, we tune the Fermi energy from 0.09 to 0.40 eV, which results in the PRF graphically displayed as colored down triangles in Figure 8 (see also Table S3 in the SI). These are obtained at the ω FQ level using the open-source plasmonX software.⁹³

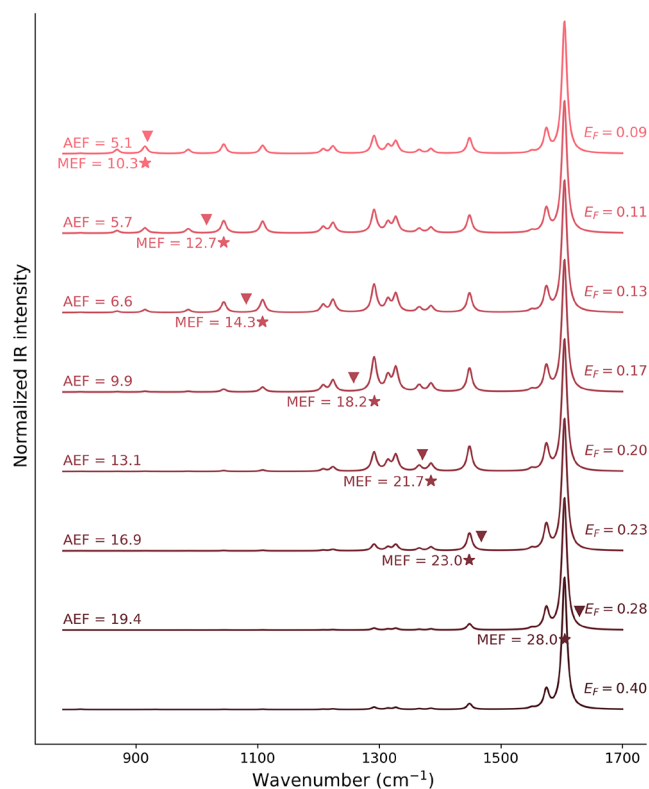


Figure 8. Normalized QM/ ω FQ SEIRA spectra of adenine in the parallel configuration on GD32 as a function of the Fermi energy (in eV). Stars denote the i -MEFs; triangles indicate the PRFs. AEF and MEF values are also reported.

As the Fermi energy decreases, the PRF of GD32 progressively redshifts, selectively activating ADE normal modes that resonate with the PRF. The i -MEF follows such redshift, while both AEF and MEF gradually decrease (as also highlighted in Table S5 in the SI). The maximum AEF (19.4) and MEF (27.9) are observed at $E_F = 0.28$ eV, which are more than five times larger than those obtained when $E_F = 0.4$ eV (AEF = 3.8 and MEF = 5.5). This is also reflected by the fact that the spectral shape gradually changes by lowering E_F , i.e., as the PRF moves toward the IR spectral range of ADE's vibrational modes. This can also be appreciated by plotting the enhancement factors for all normal modes as a function of E_F (Figure S10 in the SI). In fact, specific vibrational modes become preferentially enhanced according to their proximity to the PRF. For example, at $E_F = 0.23$ – 0.20 eV, the PRFs (1468–1371

cm^{-1}) enhance modes in the 1300–1500 cm^{-1} region, whereas at $E_F = 0.13$ – 0.09 eV (PRF 1081–919 cm^{-1}), the enhancement shifts toward lower-wavenumber modes with peaks around 900 cm^{-1} emerging distinctly. This is in perfect agreement with the behavior observed for graphene disks of increasing size, highlighting again the direct correlation between the PRF and adenine IR modes.

Interestingly, GD32 with $E_F = 0.13$ eV and GD100 with $E_F = 0.4$ eV exhibit nearly identical PRFs (1081 and 1089 cm^{-1} , respectively) with very similar AEF and MEF values (6.58 vs 7.12 and 14.30 vs 15.02), and the same i -MEF (at 1108 cm^{-1}). This highlights that, by appropriately tuning the Fermi energy of a smaller disk, the QM/ ω FQ method can effectively replicate the SEIRA spectrum of a molecule adsorbed on a larger graphene disk (GD100, containing roughly ten times more atoms), by using smaller structures (GD32).

4. CONCLUSIONS

We have presented a fully atomistic multiscale method to simulate the SEIRA spectrum of molecular systems adsorbed on plasmonic substrates. The method couples a QM description of the target molecule with a classical, fully atomistic model for the plasmonic substrate. Specifically, the atomistic electromagnetic models ω FQ and ω FQ μ describe the plasmonic behavior of noble metal nanostructures and graphene-based materials.

As a first application, we have computed the SEIRA spectra of the adenine nucleobase adsorbed on gold nanostructures, systematically analyzing the influence of molecular orientation and adsorption site on spectral features and enhancement factors. By examining 12 configurations, six on the vertex and six on the face of a icosahedral gold nanoparticle, we have directly compared SEIRA and SERS spectra and enhancements and benchmarked our results against available experimental spectra. Our analysis reveals that SERS is generally more sensitive than SEIRA, exhibiting higher enhancement factors. Nonetheless, SEIRA and SERS offer complementary spectroscopic information: while SERS is strongly influenced by the adsorption site, which dictates the spectral profile and selectively activates vibrational modes compared with the gas phase, SEIRA is particularly sensitive to the molecular binding configuration. Overall, both the adsorption site and the molecular configuration play decisive roles in shaping the spectral features and modulating enhancement in the two spectroscopies. Following the recent trend on graphene-based materials as SEIRA platforms,²² we have also explored the SEIRA spectra of ADE on graphene-based nanostructures, by systematically studying the dependence of ADE SEIRA response on the size and Fermi energy of graphene nanodisks, providing insights toward optimizing molecular detection on graphene-based SEIRA platforms.

At this stage, we can already provide a rationalization of the experimental results; however, some discrepancies are still present. These can arise from factors that are not currently captured by our current model, e.g., solvent effects and dynamical configurational phase-space sampling. This emphasizes the need for more realistic simulations, including explicit solvent molecules, pH effects, nonideal substrates, chemical effects,^{57,87} and a statistically meaningful set of configurations, potentially extracted from molecular dynamics trajectories.⁸² Addressing these aspects could, therefore, advance the predictive power of our computational strategy.

■ ASSOCIATED CONTENT

SI Supporting Information

The Supporting Information is available free of charge at <https://pubs.acs.org/doi/10.1021/acs.jpcc.5c07549>.

SEIRA dependence on DFT grid quality and differentiation step. Effect of QM/FQF μ contribution into the Hamiltonian. ADE normal modes. Raw data of enhancement factors for each ADE-substrate. SEIRA dependence on the size of Au NP. PRF values of GD as a function of the radius and Fermi energy (PDF)

■ AUTHOR INFORMATION

Corresponding Authors

Tommaso Giovannini – Department of Physics and INFN, University of Rome Tor Vergata, 00133 Rome, Italy; Consorzio Interuniversitario Nazionale per la Scienza e Tecnologia dei Materiali (INSTM), UdR Roma Tor Vergata, 00133 Rome, Italy; orcid.org/0000-0002-5637-2853; Email: tommaso.giovannini@uniroma2.it

Chiara Cappelli – Scuola Normale Superiore, Classe di Scienze, 56126 Pisa, Italy; Consorzio Interuniversitario Nazionale per la Scienza e Tecnologia dei Materiali (INSTM), UdR Pisa-SNS, 56126 Pisa, Italy; orcid.org/0000-0002-4872-4505; Email: chiara.cappelli@sns.it

Authors

Sveva Sodomaco – Scuola Normale Superiore, Classe di Scienze, 56126 Pisa, Italy; orcid.org/0009-0001-2377-774X

Piero Lafiosca – Scuola Normale Superiore, Classe di Scienze, 56126 Pisa, Italy; orcid.org/0000-0002-3967-0736

Complete contact information is available at <https://pubs.acs.org/doi/10.1021/acs.jpcc.5c07549>

Notes

The authors declare no competing financial interest.

■ ACKNOWLEDGMENTS

This work has received funding from the European Research Council (ERC) under the European Union's Horizon 2020 research and innovation programme (grant agreement No. 818064). We gratefully acknowledge the Center for High Performance Computing (CHPC) at SNS for providing the computational infrastructure.

■ REFERENCES

- (1) Lal, S.; Grady, N. K.; Kundu, J.; Levin, C. S.; Lassiter, J. B.; Halas, N. J. Tailoring plasmonic substrates for surface enhanced spectroscopies. *Chem. Soc. Rev.* **2008**, *37*, 898–911.
- (2) Yang, X.; Sun, Z.; Low, T.; Hu, H.; Guo, X.; de Abajo, F. J. G.; Avouris, P.; Dai, Q. Nanomaterial-based plasmon-enhanced infrared spectroscopy. *Adv. Mater.* **2018**, *30*, No. 1704896.
- (3) Hu, Y.; López-Lorente, A.-L.; Mizaikoff, B. Graphene-based surface enhanced vibrational spectroscopy: recent developments, challenges, and applications. *ACS Photonics* **2019**, *6*, 2182–2197.
- (4) Minopoli, A.; Acunzo, A.; Della Ventura, B.; Velotta, R. Nanostructured surfaces as plasmonic biosensors: A review. *Adv. Mater. Interfaces* **2022**, *9*, No. 2101133.
- (5) Langer, J.; Jimenez de Aberasturi, D.; Aizpurua, J.; Alvarez-Puebla, R. A.; Auguie, B.; Baumberg, J. J.; Bazan, G. C.; Bell, S. E.; Boisen, A.; Brolo, A. G.; Choo, J.; Cialla-May, D.; Deckert, V.; Fabris, L.; Faulds, K.; García de Abajo, F. J.; Goodacre, R.; Graham, D.; Haes, A. J.;

Haynes, C. L.; Huck, C.; Itoh, T.; Käll, M.; Kneipp, J.; Kotov, N. A.; Kuang, H.; Le Ru, E. C.; Kwee Lee, H.; Li, J.-F.; Yi Ling, X.; Maier, S. A.; Mayerhöfer, T.; Moskovits, M.; Murakoshi, K.; Nam, J.-M.; Nie, S.; Ozaki, Y.; Pastoriza-Santos, L.; Perez-Juste, J.; Popp, J.; Pucci, A.; Reich, S.; Ren, B.; Schatz, G. C.; Shegai, T.; Schlücker, S.; Tay, L.-L.; Thomas, K. G.; Tian, Z.-Q.; Van Duyne, R. P.; Vo-Dinh, T.; Wang, Y.; Willets, K. A.; Xu, C.; Xu, H.; Xu, Y.; Yamamoto, Y. S.; Zhao, B.; Liz-Marzán, L. M. Present and future of surface-enhanced Raman scattering. *ACS Nano* **2020**, *14*, 28–117.

(6) Papisizza, M.; Cuesta, A. In situ monitoring using ATR-SEIRAS of the electrocatalytic reduction of CO₂ on Au in an ionic liquid/water mixture. *ACS Catal.* **2018**, *8*, 6345–6352.

(7) Heidary, N.; Ly, K. H.; Kornienko, N. Probing CO₂ conversion chemistry on nanostructured surfaces with operando vibrational spectroscopy. *Nano Lett.* **2019**, *19*, 4817–4826.

(8) Alvarez-Malmagro, J.; Rueda, M.; Prieto, F. In situ surface-enhanced infrared spectroscopy study of adenine-thymine co-adsorption on gold electrodes as a function of the pH. *J. Electroanal. Chem.* **2018**, *819*, 417–427.

(9) Alvarez-Malmagro, J.; Prieto, F.; Rueda, M. In situ surface enhanced infrared absorption spectroscopy study of the adsorption of cytosine on gold electrodes. *J. Electroanal. Chem.* **2019**, *849*, No. 113362.

(10) Alvarez-Malmagro, J.; Rueda, M.; Prieto-Dapena, F. Guanine adsorption on gold electrodes studied by in situ surface-enhanced infrared reflection absorption spectroscopy. *Electrochim. Acta* **2023**, *456*, No. 142465.

(11) Quirk, A.; Unni, B.; Burgess, I. J. Surface enhanced infrared studies of 4-methoxypyridine adsorption on gold film electrodes. *Langmuir* **2016**, *32*, 2184–2191.

(12) Piergies, N.; Oćwieja, M.; Paluszkiwicz, C.; Kwiatek, W. Nanoparticle stabilizer as a determining factor of the drug/gold surface interaction: SERS and AFM-SEIRA studies. *Appl. Surf. Sci.* **2021**, *537*, No. 147897.

(13) Yamada, T.; Shirasaka, K.; Takano, A.; Kawai, M. Adsorption of cytosine, thymine, guanine and adenine on Cu (1 1 0) studied by infrared reflection absorption spectroscopy. *Surf. Sci.* **2004**, *561*, 233–247.

(14) Kundu, J.; Neumann, O.; Janesko, B.; Zhang, D.; Lal, S.; Barhoumi, A.; Scuseria, G.; Halas, N. Adenine- and adenosine monophosphate (AMP)- gold binding interactions studied by surface-enhanced Raman and infrared spectroscopies. *J. Phys. Chem. C* **2009**, *113*, 14390–14397.

(15) Rodrigo, D.; Limaj, O.; Janner, D.; Etezadi, D.; García de Abajo, F. J.; Pruneri, V.; Altug, H. Mid-infrared plasmonic biosensing with graphene. *Science* **2015**, *349*, 165–168.

(16) Wagner, M.; Seifert, A.; Liz-Marzán, L. M. Towards multi-molecular surface-enhanced infrared absorption using metal plasmonics. *Nanoscale Horiz.* **2022**, *7*, 1259–1278.

(17) Hartstein, A.; Kirtley, J.; Tsang, J. Enhancement of the infrared absorption from molecular monolayers with thin metal overlayers. *Phys. Rev. Lett.* **1980**, *45*, No. 201.

(18) Osawa, M.; Ataka, K.-I.; Ikeda, M.; Uchihara, H.; Nanba, R. Surface enhanced infrared absorption spectroscopy mechanism and application to trace analysis. *Anal. Sci.* **1991**, *7*, 503–506.

(19) Morton, S. M.; Silverstein, D. W.; Jensen, L. Theoretical studies of plasmonics using electronic structure methods. *Chem. Rev.* **2011**, *111*, 3962–3994.

(20) Osawa, M.; Ataka, K.-I.; Yoshii, K.; Nishikawa, Y. Surface-enhanced infrared spectroscopy: the origin of the absorption enhancement and band selection rule in the infrared spectra of molecules adsorbed on fine metal particles. *Appl. Spectrosc.* **1993**, *47*, 1497–1502.

(21) Neubrech, F.; Huck, C.; Weber, K.; Pucci, A.; Giessen, H. Surface-enhanced infrared spectroscopy using resonant nanoantennas. *Chem. Rev.* **2017**, *117*, 5110–5145.

(22) Oh, S.-H.; Altug, H.; Jin, X.; Low, T.; Koester, S. J.; Ivanov, A. P.; Edel, J. B.; Avouris, P.; Strano, M. S. Nanophotonic biosensors harnessing van der Waals materials. *Nat. Commun.* **2021**, *12*, No. 3824.

- (23) Kryachko, E. S.; Remacle, F. Complexes of DNA bases and gold clusters Au₃ and Au₄ involving nonconventional N-H...Au hydrogen bonding. *Nano Lett.* **2005**, *5*, 735–739.
- (24) Pan, X.; Qian, C.; Chow, A.; Wang, L.; Kamenetska, M. Atomically precise binding conformations of adenine and its variants on gold using single molecule conductance signatures. *J. Chem. Phys.* **2022**, *157*, No. 234201, DOI: 10.1063/5.0103642.
- (25) Zhang, P.; Jin, W.; Liang, W. Size-dependent optical properties of aluminum nanoparticles: from classical to quantum description. *J. Phys. Chem. C* **2018**, *122*, 10545–10551.
- (26) García de Abajo, F. J.; Howie, A. Retarded field calculation of electron energy loss in inhomogeneous dielectrics. *Phys. Rev. B* **2002**, *65*, No. 115418.
- (27) Corni, S.; Tomasi, J. Enhanced response properties of a chromophore physisorbed on a metal particle. *J. Chem. Phys.* **2001**, *114*, 3739–3751.
- (28) Corni, S.; Tomasi, J. Theoretical evaluation of Raman spectra and enhancement factors for a molecule adsorbed on a complex-shaped metal particle. *Chem. Phys. Lett.* **2001**, *342*, 135–140.
- (29) Corni, S.; Tomasi, J. Surface enhanced Raman scattering from a single molecule adsorbed on a metal particle aggregate: A theoretical study. *J. Chem. Phys.* **2002**, *116*, 1156–1164.
- (30) Corni, S.; Tomasi, J. *Surface-Enhanced Raman Scattering: Physics and Applications*; Kneipp, K.; Kneipp, H.; Moskovits, M., Eds.; Springer: Berlin/Heidelberg, 2006; Vol. 103, pp 105–124.
- (31) Takenaka, M.; Taketsugu, T.; Iwasa, T. Combined computational quantum chemistry and classical electrodynamics approach for surface enhanced infrared absorption spectroscopy. *J. Chem. Phys.* **2020**, *152*, No. 164103, DOI: 10.1063/1.5143855.
- (32) Illobre, P. G.; Lafiosca, P.; Bonatti, L.; Giovannini, T.; Cappelli, C. Mixed atomistic–implicit quantum/classical approach to molecular nanoplasmonics. *J. Chem. Phys.* **2025**, *162*, No. 044103, DOI: 10.1063/5.0245629.
- (33) Morton, S. M.; Jensen, L. A discrete interaction model/quantum mechanical method to describe the interaction of metal nanoparticles and molecular absorption. *J. Chem. Phys.* **2011**, *135*, No. 134103, DOI: 10.1063/1.3643381.
- (34) Payton, J. L.; Morton, S. M.; Moore, J. E.; Jensen, L. A hybrid atomistic electrodynamics–quantum mechanical approach for simulating surface-enhanced Raman scattering. *Acc. Chem. Res.* **2014**, *47*, 88–99.
- (35) Urbietta, M.; Barbry, M.; Zhang, Y.; Koval, P.; Sánchez-Portal, D.; Zabala, N.; Aizpurua, J. Atomic-scale lightning rod effect in plasmonic picocavities: A classical view to a quantum effect. *ACS Nano* **2018**, *12*, 585–595.
- (36) Zakomirnyi, V. I.; Rinkevicius, Z.; Baryshnikov, G. V.; Sørensen, L. K.; Ågren, H. Extended discrete interaction model: plasmonic excitations of silver nanoparticles. *J. Phys. Chem. C* **2019**, *123*, 28867–28880.
- (37) Bonatti, L.; Gil, G.; Giovannini, T.; Corni, S.; Cappelli, C. Plasmonic resonances of metal nanoparticles: Atomistic vs. continuum approaches. *Front. Chem.* **2020**, *8*, No. 340.
- (38) Pei, Z.; Mao, Y.; Shao, Y.; Liang, W. Analytic high-order energy derivatives for metal nanoparticle-mediated infrared and Raman scattering spectra within the framework of quantum mechanics/molecular mechanics model with induced charges and dipoles. *J. Chem. Phys.* **2022**, *157*, No. 164110, DOI: 10.1063/5.0118205.
- (39) Lafiosca, P.; Nicoli, L.; Bonatti, L.; Giovannini, T.; Corni, S.; Cappelli, C. QM/Classical modeling of surface enhanced Raman scattering based on atomistic electromagnetic models. *J. Chem. Theory Comput.* **2023**, *19*, 3616–3633.
- (40) Giri, S. K.; Schatz, G. C. Modeling surface-enhanced Raman scattering of Au-Pyrazine and Au-Pyrazine-Au nanorod dimer systems with the TD-DFTB method. *J. Phys. Chem. C* **2024**, *128*, 19270–19279.
- (41) Giovannini, T.; Rosa, M.; Corni, S.; Cappelli, C. A classical picture of subnanometer junctions: an atomistic Drude approach to nanoplasmonics. *Nanoscale* **2019**, *11*, 6004–6015.
- (42) Giovannini, T.; Bonatti, L.; Polini, M.; Cappelli, C. Graphene plasmonics: Fully atomistic approach for realistic structures. *J. Phys. Chem. Lett.* **2020**, *11*, 7595–7602.
- (43) Giovannini, T.; Bonatti, L.; Lafiosca, P.; Nicoli, L.; Castagnola, M.; Illobre, P. G.; Corni, S.; Cappelli, C. Do we really need quantum mechanics to describe plasmonic properties of metal nanostructures? *ACS Photonics* **2022**, *9*, 3025–3034.
- (44) Nicoli, L.; Lafiosca, P.; Grobas Illobre, P.; Bonatti, L.; Giovannini, T.; Cappelli, C. Fully atomistic modeling of plasmonic bimetallic nanoparticles: nanoalloys and core-shell systems. *Front. Photonics* **2023**, *4*, No. 1199598.
- (45) Nicoli, L.; Sodomaco, S.; Lafiosca, P.; Giovannini, T.; Cappelli, C. Atomistic Multiscale Modeling of Colloidal Plasmonic Nanoparticles. *ACS Phys. Chem. Au* **2024**, *4*, 669–678.
- (46) Bonatti, L.; Nicoli, L.; Giovannini, T.; Cappelli, C. In silico design of graphene plasmonic hot-spots. *Nanoscale Adv.* **2022**, *4*, 2294–2302.
- (47) Zanotto, S.; Bonatti, L.; Pantano, M. F.; Miseikis, V.; Speranza, G.; Giovannini, T.; Coletti, C.; Cappelli, C.; Tredicucci, A.; Toncelli, A. Strain-induced plasmon confinement in polycrystalline graphene. *ACS Photonics* **2023**, *10*, 394–400.
- (48) Giovannini, T.; Nicoli, L.; Corni, S.; Cappelli, C. The Electric Field Morphology of Plasmonic Picocavities. *Nano Lett.* **2025**, *25*, 10802–10808.
- (49) Lafiosca, P.; Giovannini, T.; Benzi, M.; Cappelli, C. Going beyond the limits of classical atomistic modeling of plasmonic nanostructures. *J. Phys. Chem. C* **2021**, *125*, 23848–23863.
- (50) Rueda, M.; Prieto, F.; Rodes, A.; Delgado, J. M. In situ infrared study of adenine adsorption on gold electrodes in acid media. *Electrochim. Acta* **2012**, *82*, 534–542.
- (51) Acres, R. G.; Cheng, X.; Beranová, K.; Bercha, S.; Skála, T.; Matolín, V.; Xu, Y.; Prince, K. C.; Tsud, N. An experimental and theoretical study of adenine adsorption on Au (111). *Phys. Chem. Chem. Phys.* **2018**, *20*, 4688–4698.
- (52) Harroun, S. G. The controversial orientation of adenine on gold and silver. *ChemPhysChem* **2018**, *19*, 1003–1015.
- (53) Hu, Y.; Chen, Q.; Ci, L.; Cao, K.; Mizaikoff, B. Surface-enhanced infrared attenuated total reflection spectroscopy via carbon nanodots for small molecules in aqueous solution. *Anal. Bioanal. Chem.* **2019**, *411*, 1863–1871.
- (54) Rodes, A.; Rueda, M.; Prieto, F.; Prado, C.; Feliu, J. M.; Aldaz, A. Adenine adsorption at single crystal and thin-film gold electrodes: an in situ infrared spectroscopy study. *J. Phys. Chem. C* **2009**, *113*, 18784–18794.
- (55) Álvarez-Malmagro, J.; Prieto, F.; Rueda, M.; Rodes, A. In situ Fourier transform infrared reflection absorption spectroscopy study of adenine adsorption on gold electrodes in basic media. *Electrochim. Acta* **2014**, *140*, 476–481.
- (56) Kundu, J.; Le, F.; Nordlander, P.; Halas, N. J. Surface enhanced infrared absorption (SEIRA) spectroscopy on nanoshell aggregate substrates. *Chem. Phys. Lett.* **2008**, *452*, 115–119.
- (57) Hu, Y.; Cao, K.; Ci, L.; Mizaikoff, B. Selective Chemical Enhancement via Graphene Oxide in Infrared Attenuated Total Reflection Spectroscopy. *J. Phys. Chem. C* **2019**, *123*, 25286–25293.
- (58) Castro Neto, A. H.; Guinea, F.; Peres, N. M.; Novoselov, K. S.; Geim, A. K. The electronic properties of graphene. *Rev. Mod. Phys.* **2009**, *81*, No. 109.
- (59) Koppens, F. H. L.; Chang, D. E.; de Abajo, F. J. G. Graphene plasmonics: a platform for strong light–matter interactions. *Nano Lett.* **2011**, *11*, 3370–3377.
- (60) Xu, H.; Xie, L.; Zhang, H.; Zhang, J. Effect of graphene Fermi level on the Raman scattering intensity of molecules on graphene. *ACS Nano* **2011**, *5*, 5338–5344.
- (61) de Abajo, F. J. G. Graphene plasmonics: challenges and opportunities. *ACS Photonics* **2014**, *1*, 135–152.
- (62) Valeš, V.; Kovaříček, P.; Fridrichová, M.; Ji, X.; Ling, X.; Kong, J.; Dresselhaus, M. S.; Kalbáč, M. Enhanced Raman scattering on functionalized graphene substrates. *2D Mater.* **2017**, *4*, No. 025087.

- (63) Pinchuk, A.; Von Plessen, G.; Kreibig, U. Influence of interband electronic transitions on the optical absorption in metallic nanoparticles. *J. Phys. D: Appl. Phys.* **2004**, *37*, No. 3133.
- (64) Pinchuk, A.; Kreibig, U.; Hilger, A. Optical properties of metallic nanoparticles: influence of interface effects and interband transitions. *Surf. Sci.* **2004**, *557*, 269–280.
- (65) Balamurugan, B.; Maruyama, T. Evidence of an enhanced interband absorption in Au nanoparticles: size-dependent electronic structure and optical properties. *Appl. Phys. Lett.* **2005**, *87*, No. 143105, DOI: 10.1063/1.2077834.
- (66) Liebsch, A. Surface-plasmon dispersion and size dependence of Mie resonance: silver versus simple metals. *Phys. Rev. B* **1993**, *48*, No. 11317.
- (67) Mayer, A. Formulation in terms of normalized propagators of a charge-dipole model enabling the calculation of the polarization properties of fullerenes and carbon nanotubes. *Phys. Rev. B Condens. Matter* **2007**, *75*, No. 045407.
- (68) Giovannini, T.; Puglisi, A.; Ambrosetti, M.; Cappelli, C. Polarizable QM/MM approach with fluctuating charges and fluctuating dipoles: the QM/FQFμ model. *J. Chem. Theory Comput.* **2019**, *15*, 2233–2245.
- (69) Cammi, R.; Cappelli, C.; Corni, S.; Tomasi, J. On the calculation of infrared intensities in solution within the polarizable continuum model. *J. Phys. Chem. A* **2000**, *104*, 9874–9879.
- (70) Cammi, R.; Mennucci, B.; Tomasi, J. On the calculation of local field factors for microscopic static hyperpolarizabilities of molecules in solution with the aid of quantum-mechanical methods. *J. Phys. Chem. A* **1998**, *102*, 870–875.
- (71) Tomasi, J.; Cammi, R.; Mennucci, B. Medium effects on the properties of chemical systems: An overview of recent formulations in the polarizable continuum model (PCM). *Int. J. Quantum Chem.* **1999**, *75*, 783–803.
- (72) Ataka, K.; Heberle, J. Biochemical applications of surface-enhanced infrared absorption spectroscopy. *Anal. Bioanal. Chem.* **2007**, *388*, 47–54.
- (73) Barhoumi, A.; Zhang, D.; Tam, F.; Halas, N. J. Surface-enhanced Raman spectroscopy of DNA. *J. Am. Chem. Soc.* **2008**, *130*, 5523–5529.
- (74) Brown, K. A.; Park, S.; Hamad-Schifferli, K. Nucleotide-surface interactions in DNA-modified Au-nanoparticle conjugates: Sequence effects on reactivity and hybridization. *J. Phys. Chem. C* **2008**, *112*, 7517–7521.
- (75) Liu, J. Adsorption of DNA onto gold nanoparticles and graphene oxide: Surf. Sci. and applications. *Phys. Chem. Chem. Phys.* **2012**, *14*, 10485–10496.
- (76) SCM, ADF .x, Theoretical Chemistry, Vrije Universiteit Amsterdam. 2020 <https://www.scm.com>. (last access December 19, 2025).
- (77) Baerends, E. J.; Aguirre, N. F.; Austin, N. D.; Autschbach, J.; Bickelhaupt, F. M.; Bulo, R.; Cappelli, C.; van Duin, A. C.; Egidi, F.; Fonseca Guerra, C.; Förster, A.; Franchini, M.; Goumans, T. P. M.; Heine, T.; Hellström, M.; Jacob, C. R.; Jensen, L.; Krykunov, M.; van Lenthe, E.; Michalak, A.; M, M. M.; Neugebauer, J.; Nicu, V. P.; Philipsen, P.; Ramanantoanina, H.; Rüger, R.; Schreckenbach, G.; Mauro, S.; Swart, M.; Thijssen, J. M.; Trnka, T.; Visscher, L.; Yakovlev, A.; van Gisbergen, S. The Amsterdam Modeling Suite. *J. Chem. Phys.* **2025**, *162*, No. 162501, DOI: 10.1063/5.0258496.
- (78) Martín Sabanés, N.; Ohto, T.; Andrienko, D.; Nagata, Y.; Domke, K. F. Electrochemical TERS Elucidates Potential-Induced Molecular Reorientation of Adenine/Au (111). *Angew. Chem.* **2017**, *129*, 9928–9933.
- (79) Yoshimoto, T.; Seki, M.; Okabe, H.; Matsuda, N.; Wu, D.-y.; Futamata, M. Three distinct adsorbed states of adenine on gold nanoparticles depending on pH in aqueous solutions. *Chem. Phys. Lett.* **2022**, *786*, No. 139202.
- (80) Prieto, F.; Su, Z.; Leitch, J. J.; Rueda, M.; Lipkowski, J. Quantitative subtractively normalized interfacial Fourier transform infrared reflection spectroscopy study of the adsorption of adenine on Au (111) electrodes. *Langmuir* **2016**, *32*, 3827–3835.
- (81) Rosa, M.; Corni, S.; Di Felice, R. Enthalpy–entropy tuning in the adsorption of nucleobases at the Au (111) surface. *J. Chem. Theory Comput.* **2014**, *10*, 1707–1716.
- (82) Sodomaco, S.; Gómez, S.; Giovannini, T.; Cappelli, C. Computational Insights into the Adsorption of Ligands on Gold Nanosurfaces. *J. Phys. Chem. A* **2023**, *127*, 10282–10294.
- (83) Jackson, J. D. *Classical electrodynamics*; John Wiley & Sons, 1999.
- (84) Mathlouthi, M.; Seuvre, A.-M.; Koenig, J. L. Ft-ir and laser-Raman spectra of adenine and adenosine. *Carbohydr. Res.* **1984**, *131*, 1–15.
- (85) Kelly, K. L.; Coronado, E.; Zhao, L. L.; Schatz, G. C. The optical properties of metal nanoparticles: the influence of size, shape, and dielectric environment. *J. Phys. Chem. B* **2003**, *107*, 668–677.
- (86) Willets, K. A.; Van Duyne, R. P. Localized surface plasmon resonance spectroscopy and sensing. *Annu. Rev. Phys. Chem.* **2007**, *58*, 267–297.
- (87) Hu, Y.; López-Lorente, Á.; Mizaikoff, B. Versatile analytical platform based on graphene-enhanced infrared attenuated total reflection spectroscopy. *ACS Photonics* **2018**, *5*, 2160–2167.
- (88) Georgiou, K.; Wang, Y.; Ma, X. Orientation Sensitive SEIRA Sensors Based on Single-Walled Carbon Nanotube Near Fields. *Nano Lett.* **2024**, *24*, 10540–10546.
- (89) Antony, J.; Grimme, S. Structures and interaction energies of stacked graphene–nucleobase complexes. *Phys. Chem. Chem. Phys.* **2008**, *10*, 2722–2729.
- (90) Voloshina, E.; Mollenhauer, D.; Chiappisi, L.; Paulus, B. Theoretical study on the adsorption of pyridine derivatives on graphene. *Chem. Phys. Lett.* **2011**, *510*, 220–223.
- (91) Berland, K.; Chakarova-Käck, S. D.; Cooper, V. R.; Langreth, D. C.; Schröder, E. A van der Waals density functional study of adenine on graphene: single-molecular adsorption and overlayer binding. *J. Phys.: Condens. Matter* **2011**, *23*, No. 135001.
- (92) Huang, S.; Ling, X.; Liang, L.; Song, Y.; Fang, W.; Zhang, J.; Kong, J.; Meunier, V.; Dresselhaus, M. S. Molecular selectivity of graphene-enhanced Raman scattering. *Nano Lett.* **2015**, *15*, 2892–2901.
- (93) Giovannini, T.; Illobre, P. G.; Lafiosca, P.; Nicoli, L.; Bonatti, L.; Corni, S.; Cappelli, C. plasmonX: an Open-Source Code for Nanoplasmonics. 2025 arXiv:2510.12731. arXiv.org e-Printarchive. <https://arxiv.org/abs/2510.12731>.